Correlation of Nanogap Formation and Strain Distribution in Single-Layer CVD MoS₂

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ABSTRACT

Molybdenum disulfide (MoS₂), a promising semiconducting material with high carrier mobility and atomic flatness, has shown potential in electronic applications. However, strain induced by the thermal expansion mismatch between MoS₂ and its growth substrate during the chemical vapor deposition (CVD) process can lead to nanogap formation. affecting its performance functionality. In this study, the correlation between strain distribution and nanogap formation in singlelayer CVD MoS₂ was investigated using photoluminescence (PL) mapping. We successfully imaged the strain distribution in MoS₂ flakes using PL mapping, and it was revealed that the adhesion between MoS₂ and substrate significantly affects strain distribution in MoS2 flakes. In addition, we can conclude that the nanogap formation can partially release the growth-induced strain in MoS₂ flakes. The nanogap propagation mechanism was also revealed by comparing the strain distribution of two MoS₂ flakes with different nanogap structures.

KEY WORDS

Molybdenum disulfide (MoS₂), photoluminescence (PL) spectroscopy, strain, nanogap, chemical vapor deposition (CVD)

1. INTRODUCTION

Molybdenum disulfide (MoS₂) is a compromising semiconducting material due to its atomic flatness and high carrier mobility. In particular, chemical vapor deposition (CVD)-grown MoS₂ has been widely explored for electronic applications due to its high quality and scalability. However, the thermal expansion coefficient mismatch between MoS₂ and growth substrate induces strain in MoS₂ flakes, and the nanogap structure can be formed to release such growth-induced strain. Analyzing and controlling this unique nanogap structure is worthwhile since it can be an intriguing opportunity to explore new functional applications, such as nanogap electrodes, biosensing devices, and gas sensors. In this work, the correlation between the strain distribution and nanogap formation in singlelayer CVD MoS₂ was investigated, which is necessary to reveal the nanogap formation mechanism and control the nanogap structure during synthesis. We successfully imaged the strain distribution MoS_2 flakes using in photoluminescence (PL) mapping, and it was revealed that the adhesion between MoS2 and substrate significantly affects strain distribution in

MoS₂ flakes. In addition, by comparing the strain distribution of the flakes with and without nanogap, it can be concluded that the formation of nanogap can partially release the growth-induced strain in MoS₂ flakes. The nanogap propagation mechanism was also revealed by comparing the strain distribution of two MoS₂ flakes with different nanogap structures.

2. RESULTS

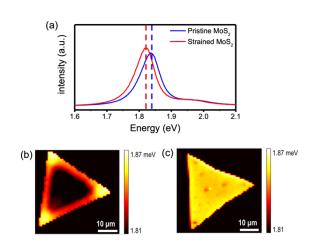


Figure 1. (a) Photoluminescence (PL) spectrum of the pristine (red line) and strained (blue line) single-layer MoS₂. PL mapping image of the single layer MoS₂ flake, which has (b) good and (c) bad adhesion with the growth substrate.

It is known that the PL peak position of single-layer MoS₂ shifts to the lower energy due to the strain in MoS₂, as shown in Figure 1(a). Therefore, a PL mapping image can visualize the strain distribution in MoS₂ flakes. Figure 1(b) and (c) are PL mapping images of the single-layer MoS₂ flake, which has good and bad adhesion with the substrate. Unlike the homogeneous signals in the bad adhesion sample, the good adhesion sample clearly shows the strain distribution. In the good adhesion sample, the relaxation of thermal strain induced in MoS₂ is retarded due to the relatively good adhesion between MoS₂ and the SiO₂ substrate during the cooling process. However, in the bad adhesion sample, the interface and edge bonding between the MoS₂ layer and the substrate are not sufficiently strong to retain the strain, resulting in the complete relaxation of thermal strain during the cooling process.

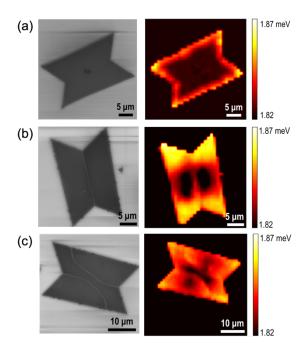


Figure 2. (a) Scanning electron microscopy (SEM) image (left) and PL mapping image of the polycrystalline MoS₂ flake without nanogap. (b) SEM image (left) and PL mapping image of the polycrystalline MoS₂ flake with nanogap following grain boundary. (c) SEM image (left) and PL mapping image of the polycrystalline MoS₂ flake with randomly oriented nanogap.

Figures 2(a), 2(b), and 2(c) are the scanning electron microscopy (SEM) (left) and PL mapping (right) images of the polycrystalline MoS₂ flakes without nanogap, with nanogap following grain boundary and randomly oriented nanogap, respectively. In the PL mapping image, the MoS₂ flake without nanogap shows a consistent peak position across the flake except for the edge area where partial strain release occurs (Figure 2(a)). In contrast, MoS₂ flakes with nanogap exhibit a peak position shift towards higher energies along the propagation direction of the nanogap, indicating a partial release of growth-induced strain along the path of the nanogap formation (Figures 2(b) and 2(c)). This result suggests that the thermal strain in MoS₂ flakes can be partially released by forming the nanogap structure. Moreover, the direction of nanogap propagation can also be related to the adhesion of MoS₂ and the substrate. Nanogap can be formed following the grain boundary of the polycrystalline MoS₂ flakes, as shown in Figure 2(b). However, nanogaps can be formed randomly, as shown in Figure 2(c). In 3 different samples that have different adhesion properties with the substrate, the ratio of the number of flakes with nanogap following the grain boundary and flakes with nanogap both following the grain boundary and random direction are calculated, and it was confirmed that the better the adhesion, the higher the ratio (47.8%, 36.3% and 12.1%). Due to the weak interaction with the underlying substrate, the bad adhesion sample can undergo higher mechanical deformation than the good adhesion

sample. Therefore, in the bad adhesion sample, nanogap formation mainly occurred in random directions, starting from the edges of the grain boundary. However, in the good adhesion sample, sufficient interface bonding between the flakes and the substrate can make nanogap propagation occur at a relatively low speed, resulting in a higher ratio of nanogap formation along the grain boundary than in the case of the bad adhesion sample.

3. CONCLUSION

The strain distribution in various MoS₂ flakes was successfully measured using PL mapping. Adhesion between the MoS₂ layer and the substrate plays a key role in strain distributions of MoS2 flakes and the nanogap formation occurs to release the strain and the strain distribution can affect to resulting structure of the nanogap.

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